

SHELXT: Integrating space group determination and structure solution

(also known as XT or Intrinsic phasing)

ACA Summer School, June 23rd 2015

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<http://shelx.uni-ac.gwdg.de/SHELX/>

Introduction

SHELXT is a program for solving relatively routine small molecule structures, determining the space group and structure together. Most of the methods it uses are well established, but they are combined in a way that is designed to be robust, efficient and very easy to use.

SHELXT reads standard SHELX format *name.ins* and *name.hkl* files. The CELL, LATT, SYMM, SFAC and HKLF cards are read, but the input space group and the rest of the *.ins* file are ignored. The Laue group is extracted but may be overridden with the `-L` command line switch, e.g. `-L15` to make the program try all trigonal and hexagonal Laue groups. Other command line options are possible but are rarely needed; for a full list, run SHELXT without a filename. SHELXT is a stand-alone executable, no other files, DLLs, environment variables etc. are needed. To solve a structure enter:

`shelxt name`

Space groups for small and macro-molecules

For data collection and scaling, we must first establish the Laue group and lattice type. To keep it simple, let us assume that we have an orthorhombic crystal, i.e. Laue group mmm , and a primitive lattice.

If it is a protein, there will be 4 possible space groups – $P2_12_12_1$ (23.9%), $P2_12_12$ (6.1%), $P222_1$ (0.04%) and $P222$ (0.01%). Note the very unequal distribution in the PDB! We can simply ask a MR program such as PHASER to try all 8 (!) possibilities, and if only one leads to a solution, then it is the correct space group and orientation. Since three-quarters of proteins are solved by MR anyway, this is a popular and reliable method of determining their space groups.

If it is a small molecule, there are 30 orthorhombic-P space groups and 111 different possible combinations of space group and orientation for them, so it is more interesting!

Small molecule example: $[\text{Co}(\text{OH}_2)_6]^{2+} [\text{NH}_4]^+ [\text{PO}_4]^{3-}$

This example is taken from Peter Müller's book (OUP, 2006). XPREP looks for the highest metric symmetry, in this case orthorhombic P, and sets $a < b < c$. The low R(sym) confirms Laue symmetry mmm.

Determination of reduced (Niggli) cell

Transformation from original cell (HKLF-matrix):

```
-1.0000  0.0000  0.0000   0.0000  0.0000 -1.0000   0.0000 -1.0000  0.0000
```

Unitcell: 6.112 6.941 11.196 90.00 90.00 90.00

Niggli form: a.a = 37.36 b.b = 48.18 c.c = 125.35
 b.c = 0.00 a.c = 0.00 a.b = 0.00

Search for higher METRIC symmetry

Identical indices and Friedel opposites combined before calculating R(sym)

```
-----  
Option A: FOM = 0.000 deg.    ORTHORHOMBIC P-lattice    R(sym) = 0.022 [ 1607]  
Cell:     6.112     6.941     11.196     90.00     90.00     90.00     Volume:            474.97  
Matrix:-1.0000  0.0000  0.0000  0.0000  0.0000 -1.0000  0.0000 -1.0000  0.0000  
-----
```

Option B retains original cell

Select option [A]:

XPREP space group determination

XPREP looks for space groups that fit the Laue group, lattice type, systematic absences, $\langle |E^2-1| \rangle$ and frequency of the space group in the CSD. The last two have lower weight in the CFOM. A and B have different axial orientations but the same absences and CFOM, so XPREP cannot distinguish them. $\langle |E^2-1| \rangle$ indicates non-centrosymmetric (or twinning).

Mean $|E^2-1| = 0.677$ [expected .968 centrosym and .736 non-centrosym]

Systematic absence exceptions:

	b--	c--	n--	21--	-c-	-a-	-n-	-21-	--a	--b	--n	--21
N	508	508	580	22	438	445	429	29	319	316	311	39
N I>3s	205	205	0	19	391	435	384	0	302	276	264	0
<I>	463.4	464.0	3.1	895.3	565.6	621.6	568.7	2.8	892.6	542.6	833.4	2.3
<I/s>	10.4	10.4	0.4	15.3	16.3	19.4	16.4	0.4	17.7	15.8	14.8	0.5

Identical indices and Friedel opposites combined before calculating R(sym)

Option	Space Group	No.	Type	Axes	CSD	R(sym)	N(eq)	Syst. Abs.	CFOM
[A]	Pmn2(1)	# 31	non-cen	2	53	0.022	1607	0.5 / 10.4	3.20
[B]	Pmn2(1)	# 31	non-cen	5	53	0.022	1607	0.5 / 10.4	3.20
[C]	Pmnn	# 59	centro	3	42	0.022	1607	0.5 / 10.4	11.80

Select option [B]:

Structure solution with XT/SHELXT

SHELXT finds *one* solution. Both R1 and α (normalized mean square phase error) are convincing. The Flack x may improve after anisotropic refinement, but still gets the correct absolute structure. The figure of merit Chem is only meaningful for organic structures, for which it should be close to 1. All atoms are correctly assigned. Note that a change in cell orientation is also involved. The total time on an i7 desktop was less than 1 second.

Try	N(iter)	CC	R(weak)	Chem	CFOM	best	Sig(min)	N(P1)	Vol/N
1	100	94.57	0.0220	0.0000	0.9238	0.9238	4.219	26	18.27
2	100	93.98	0.0278	0.0000	0.9120	0.9238	2.422	27	17.59
3	100	94.29	0.0249	0.0000	0.9180	0.9238	2.362	27	17.59
4	100	94.20	0.0250	0.0000	0.9170	0.9238	2.507	27	17.59

4 attempts, solution 1 selected with best CFOM = 0.9238, Alpha0 = 0.343

Structure solution: 0.381 secs

0 Centrosymmetric and 56 non-centrosymmetric space groups evaluated

Space group determination: 0.341 secs

R1	Rweak	Alpha	Orientation	Space group	Flack_x	File	Formula
0.033	0.016	0.000	a'=c, b'=a, c'=b	Pmn2(1)	0.23	t071_a	O10 P Co N

```

TITL t071_a.res in Pmn2(1)
REM Old TITL [NH4]+ [Co(H2O)6]2+ [PO4]3- from Peter Müller
REM SHELXT solution in Pmn2(1)
REM R1 0.033, Rweak 0.016, Alpha 0.000, Orientation a'=c, b'=a, c'=b
REM Flack x = 0.226 ( 0.012 ) from Parsons' quotients
REM Formula found by SHELXT: O10 P Co N
CELL 0.71073 6.9410 6.1120 11.1960 90.000 90.000 90.000
ZERR 2.000 0.0020 0.0020 0.0030 0.000 0.000 0.000
LATT -1
SYMM 1/2-X, -Y, 1/2+Z
SYMM 1/2+X, -Y, 1/2+Z
SYMM -X, Y, Z
SFAC O H P CO N
UNIT 20 32 2 2 2
L.S. 10
BOND $H
LIST 6
FMAP 2
PLAN 20
CO01 4 0.50000 0.86717 0.47826 10.50000 0.01068 27.94
P002 3 1.00000 0.49687 0.60602 10.50000 0.01000 16.12
O003 1 0.70713 0.97162 0.59805 11.00000 0.01877 7.75
O004 1 1.00000 0.51268 0.47027 10.50000 0.01200 7.73
O005 1 1.00000 0.72904 0.65992 10.50000 0.01325 7.64
O006 1 0.27539 0.76252 0.36826 11.00000 0.01556 7.59
O007 1 0.81857 0.37423 0.64765 11.00000 0.01396 7.51
O008 1 0.50000 0.54957 0.55993 10.50000 0.01833 7.12
O009 1 0.50000 1.18203 0.39532 10.50000 0.02240 6.95
N00A 5 1.00000 0.12487 0.34008 10.50000 0.01931 5.97
HKLF 4
END

```

Output file from SHELXT (t071_a.res)

Electron count



CO01	4	0.50000	0.86717	0.47826	10.50000	0.01068	27.94
P002	3	1.00000	0.49687	0.60602	10.50000	0.01000	16.12
O003	1	0.70713	0.97162	0.59805	11.00000	0.01877	7.75
O004	1	1.00000	0.51268	0.47027	10.50000	0.01200	7.73
O005	1	1.00000	0.72904	0.65992	10.50000	0.01325	7.64
O006	1	0.27539	0.76252	0.36826	11.00000	0.01556	7.59
O007	1	0.81857	0.37423	0.64765	11.00000	0.01396	7.51
O008	1	0.50000	0.54957	0.55993	10.50000	0.01833	7.12
O009	1	0.50000	1.18203	0.39532	10.50000	0.02240	6.95
N00A	5	1.00000	0.12487	0.34008	10.50000	0.01931	5.97

The P1 approach

It is well established that most direct methods often work best if the data are first expanded to space group P1 [Sheldrick & Gould, *Acta Cryst.* B51 (1995) 423-431] . This suggests the following approach:

1. Assuming the Laue group to be known, equivalent intensities are averaged and the data then expanded to P1.
2. The phase problem is solved by dual space recycling in P1. The result is an electron density map and the corresponding phases.
3. The *phases* are used to determine the correct space group and the translation necessary for the electron density map to fit it.
4. The phases are averaged in this space group and used to calculate an improved density that is improved further by Fourier recycling.
5. The maxima of the density are assigned to atoms.

Starting from the Patterson function

Unlike almost all direct methods of the last three decades, SHELXT is (almost) *deterministic*. Although dual-space direct methods usually start from random phases or atoms, in the presence of some heavier atoms a considerable speed-up can be achieved by starting from a Patterson superposition minimum function.

If two copies of the Patterson, displaced from one another by a vector corresponding to a strong Patterson peak, are superimposed and their minimum function calculated, it should correspond to a double image of the structure with shifted origins in the effective space group P1.

So if we are planning to solve the structure first in P1 anyway, this is an excellent starting point, and should never be worse than starting from random phases.

Random OMIT maps

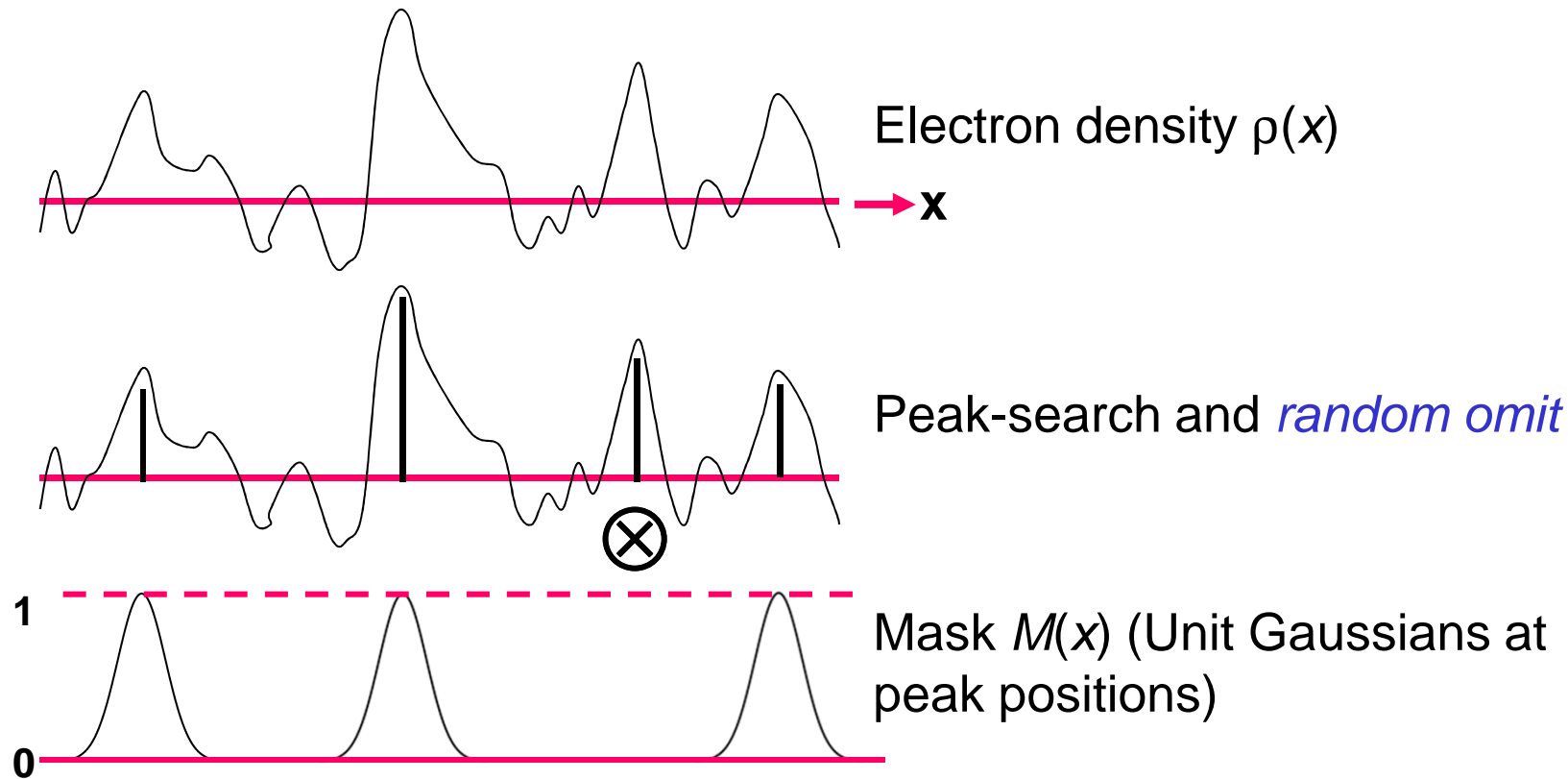
Omit maps were frequently used by protein crystallographers to reduce *model bias* when interpreting unclear regions of a structure. A small part (<10%) of the model is deleted, then the rest of the structure refined (often with simulated annealing to reduce memory effects) and finally a new difference electron density map is calculated.

A key feature of SHELXD when it is used to solve large equal-atom structures is the use of *random omit maps*. About 30% of the peaks are omitted at random each cycle and the remaining atoms used to calculate phases. This can be regarded as a *perturbation* of the density, like charge flipping.

Random omit maps are also employed in the dual-space recycling part of SHELXT (making it somewhat less deterministic).

Random omit without atoms

The following dual space algorithm enables the *random omit* method to be combined efficiently with FFTs in both directions. It achieves similar results to charge flipping, but imposes more atomicity.



The new density is calculated as $\rho'(x) = \rho(x) \cdot M(x)$. This also truncates negative density and sharpens the remaining atoms!

A phase determination strategy

1. Generate starting maps in P1 by Patterson superposition. This of necessity produces a trial structure in P1.
2. Iterate using the *random omit mask* perturbation. Use sharpened amplitudes $G_{hkl} = (EF)^{1/2} \{ F_o^2 / [F_o^2 + \sigma^2(F_o)] \}$ and Fourier coefficients $3G_o - 2G_c$ with phases ϕ_c .
3. The phases with the best CFOM are used to find the space group followed by further density modification. **CFOM = 0.01CC – R(weak)**, where R(weak) is the mean value of E_c^2 for reflections with smallest E_o^2 .

Unlike most other direct methods, this algorithm works reasonably well with heavy atoms on special positions and for pseudo-symmetric structures. Because it indirectly assumes *atoms* it is less demanding on data quality and completeness than charge flipping, but charge flipping should work better for modulated and severely disordered structures, because it *doesn't assume atoms*.

Using phases to find the correct origin and space group

PLATON can find the space group and place its origin correctly starting from atom positions in P1. The disadvantage of this approach is that tolerances are required to decide whether atom positions are the same within experimental error in the higher space group.

Giacovazzo [*J. Appl. Cryst* 33 (2000) 307] and Palatinus [*J. Appl. Cryst.* 41 (2008) 975] suggested ways of finding the space group and the required origin shift using only the phases. The two approaches are similar but Giacovazzo finds the full space group directly (but at his own admission, slowly) whereas Palatinus searches for individual symmetry elements and then uses them to construct the space group.

An advantage of this approach is that it is possible to define a single figure of merit to decide which possible space group is more likely.

The phases of equivalent reflections

The phase ϕ_m of the equivalent reflection \mathbf{h}_m is derived from the phase ϕ of the (prime) reflection \mathbf{h} by:

$$\phi_m = \phi - 2\pi \mathbf{h} \mathbf{t}_m = \phi - 2\pi (h t_1 + k t_2 + \ell t_3)$$

For example in $P3_1$:
(for symmetry operator #2)

$$h_2 = 0h + 1k + 0\ell = k$$

$$k_2 = -1h - 1k + 0\ell = -h - k$$

$$\ell_2 = 0h + 0k + 1\ell = \ell$$

So \mathbf{h}_2 is $k, -h-k, \ell$ with phase:

$$\phi_2 = \phi - 2\pi \mathbf{h} \mathbf{t}_2 = \phi - 2\pi (0h + 0k + \frac{1}{3}\ell) = \phi - (\frac{2}{3})\pi\ell$$

This relation is only valid if the space group is correct and the structure has been shifted so that the origin is correctly placed.

Phases and translation

If the whole structure is shifted by $\Delta\mathbf{x}$ we can write: $\mathbf{x}_j' = \mathbf{x}_j + \Delta\mathbf{x}$

$$\begin{aligned}\text{Thus } \mathbf{F}_{\mathbf{h}'} &= \sum_j f_j \exp(2\pi i \mathbf{h}\mathbf{x}'_j) = \sum_j f_j \exp(2\pi i (\mathbf{h}(\mathbf{x}_j + \Delta\mathbf{x}))) \\ &= \left[\sum_j f_j \exp(2\pi i \mathbf{h}\mathbf{x}_j) \right] \exp(2\pi i \mathbf{h}\Delta\mathbf{x}) \\ &= \mathbf{F}_{\mathbf{h}} \exp(2\pi i \mathbf{h}\Delta\mathbf{x})\end{aligned}$$

i.e. $\phi' = \phi + 2\pi\mathbf{h}\Delta\mathbf{x}$

This provides us with a way to find the origin shift $\Delta\mathbf{x}$ using the phases, by fitting the phase differences between equivalent reflections \mathbf{h}' and \mathbf{h} to $2\pi\mathbf{h}\Delta\mathbf{x}$. This calculation has to be performed modulo 2π !

The α figure of merit

In my hands both the Giacovazzo and Palatinus figures of merit performed well. The related α value used in SHELXT has a direct physical meaning (the normalized mean square phase error). For a prime reflection \mathbf{h} and a symmetry equivalent \mathbf{h}_m we define:

$$q = \{ \phi_m - \phi + 2\pi[\mathbf{h}\mathbf{t}_m + \Delta\mathbf{x}(\mathbf{h}_m - \mathbf{h})] \} \text{ modulo } 2\pi$$

For the correct origin shift $\Delta\mathbf{x}$ and the correct space group q should be close to zero if the phases of the symmetry equivalents are consistent. An F^2 -weighted sum of q^2 over all pairs of equivalents for all reflections, normalized so that it would have a value of 1.0 for random phases, is then the figure of merit α .

Finding the origin in a centrosymmetric space group

Only when the inversion center of a centrosymmetric structure is at the origin will the phases be 0 or π , otherwise they have general values. To find the origin shift needed to bring an inversion center to the origin, we can double the phases (so that they should all be zero if the origin is correct) and then perform a Fourier transformation:

$$P_X = \sum_h |F_h|^2 \exp(-2\pi i \cdot 2\phi_h) \exp(-2\pi i \cdot hX)$$

The electron density should then be shifted by $X/2$ to bring it to the true origin, where X is the position of the maximum of this *origin shifted Patterson function*. In SHELXT the α figure of merit for this P1 to P $\bar{1}$ conversion is referred to as α_0 . It should be less than about 0.25 for a centrosymmetric space group.

It is still necessary to take into account that not all inversion centers are equivalent in all space groups. However it is still much faster to test all possible non-equivalent inversion centers than to do a full 3D grid search to find the best value of α .

Calculating α in non-centrosymmetric space groups

A 3D grid search for α would be slow because it is not suitable for a FFT, so it is divided into 2D and 1D searches. All non-centrosymmetric space groups in the given Laue group are tested as follows, taking axis transformations to obtain conventional settings into account where necessary.

1. For P1 no search is required, α is undefined (set to zero).
2. For the space groups Pm, Pc, Pn, Cm and Cc a 1D line search is performed.
3. For all other polar space groups, a 2D grid search is performed.
4. For all other space groups, a 2D grid search is followed by a 1D line search.

In all cases (including centrosymmetric) the Δx value obtained by interpolation is then refined further to minimize α .

Preliminary element assignment

In each of the possible space groups, if necessary after re-orientating the axes to obtain the conventional setting, further dual-space recycling is performed to improve the quality of the electron density. The peaks in this map are integrated to get electron counts and these are used to assign atoms, assuming that all possible elements present have been specified (SFAC is used but not UNIT).

There is a problem in putting the electron counts onto an absolute scale. Currently this is solved by looking for some typical organic junk and assuming that it is carbon, or for inorganics by looking for typical groups such as oxyanions. If this cannot be done, the program assumes that the atom with the highest density corresponds to the element with the highest atomic number on the SFAC instruction.

At this point some simple chemical rules are applied to avoid nonsense assignments. If one or more heavy atoms are clearly present but not given on SFAC, the program tries Cl, Br or I.

The free lunch algorithm

If the data are incomplete, SHELXT can simply invent the data that it would have liked to have but doesn't (the *free lunch algorithm*, Caliandro *et al.*, (2005). *Acta Cryst.* **D61**, 556-565). This can be useful in the following cases :

1. It was not possible to measure complete data, e.g. because a high-pressure cell was used.
2. To check what the compound is (and possible reject the crystal) before data collection is complete.
3. To obtain more complete structures of poorly diffracting crystals by artificially extending the resolution.
4. The electron density integration used in the element assignment works better if strong but missing low order reflections are included in this way.

Isotropic refinement and the Flack x parameter

After assigning the atoms, an isotropic refinement is performed and R1 calculated. Atoms with very high U-values are eliminated after the isotropic refinement but to save time no further refinement is performed. This may well change in future versions.

For non-centrosymmetric space groups, a Flack x parameter is estimated by the Parsons' quotient algorithm. If x is greater than 0.5 the coordinates and if necessary the space group are inverted. *Thus the structure determined by SHELXT almost always has the correct hand!*

x can also be good indication as to whether the space group is correct. However the value so obtained is not as good as the value after the final refinement.

Bringing the atoms together

The UNIQ instruction in XP was the standard for building molecules, but the following algorithm used in SHELXT is better, because it does not require that elements and hence covalent radii are correctly assigned to the atoms!

1. Generate the SDM (Shortest Distance Matrix – shortest distances between unique atoms, taking symmetry into account).
2. Set a flag to -1 for each unique atom, then change it to $+1$ for one atom - it does not matter which.
3. Search the SDM for the shortest distance for which the product of the two flags is -1 ; *if none, exit*.
4. Symmetry transform the atom with flag -1 for this distance so that it is as close as possible to the atom with flag $+1$, then change its flag to $+1$.
5. GOTO 3

This diabolically simple algorithm not only builds the molecules as we would intuitively expect them whatever the space group, but also clusters them in a chemically sensible way, making the structure instantly recognizable. SHELXT also centers the structure in the unit-cell.

Pseudosymmetry

When the space group is centrosymmetric but the heavy atom substructure is non-centrosymmetric, there can be many apparent solutions. Sometimes the result is a mess. For this Pt complex, R1 and x indicate the correct model.

R1	Rweak	Alpha	Space group	Flack_x	File	Formula
0.634	0.338	0.051	P4/mmm		t240_a	C35 N6 Pt96 I
0.203	0.004	0.069	P4(2)/mmc		t240_b	C42 N12 F6 Cl8 Pt
0.637	0.398	0.070	P4(2)/mnm		t240_c	C22 N8 Pt46 I
0.208	0.007	0.071	P4/mnc		t240_d	C38 N26 F14 Cl4 Pt
0.552	0.137	0.035	P-4m2	0.48	t240_e	C59 N2 F8 Cl10 Pt67
0.165	0.004	0.051	P4mm	0.50	t240_f	C83 N8 F25 Cl16 Pt2
0.168	0.004	0.053	P422	0.50	t240_g	C39 N16 F14 Cl5 Pt
0.181	0.004	0.053	P-42m	0.49	t240_h	C46 N16 F44 Cl12 Pt I
0.146	0.003	0.054	P4(2)mc	0.49	t240_i	C41 N5 F5 Cl2 Pt
0.123	0.003	0.055	P4(2)2(1)2	0.09	t240_j	C30 N6 F14 Cl2 Pt
0.139	0.003	0.056	P-42(1)c	0.47	t240_k	C26 N16 F12 Pt
0.157	0.004	0.072	P-42(1)m	0.49	t240_l	C41 N14 F12 Cl3 Pt
0.515	0.120	0.072	P-42c	0.46	t240_m	C16 Pt68 I
0.529	0.146	0.073	P4(2)22	0.47	t240_n	C20 N4 Pt58 I
0.166	0.004	0.073	P42(1)2	0.44	t240_o	C48 N21 F5 Cl4 Pt
0.142	0.006	0.074	P4(2)nm	0.50	t240_p	C42 N22 F12 Cl2 Pt
0.192	0.007	0.075	P4nc	0.50	t240_q	C41 N22 F14 Cl9 I
0.164	0.007	0.076	P-4n2	0.49	t240_r	C42 N22 F14 Cl4 Pt

Both the crystallographers and SHELXT can fail!

R1	Rweak	Alpha	Orientation	Space group	Flack_x	File	Formula
0.073	0.005	0.018	as input	P6(3)/mmc		i027_a	Ca4 O17 S2
0.074	0.004	0.016	as input	P6(3)mc	no Fp	i027_b	Ca O17 S5
0.039	0.004	0.017	as input	P-6m2	no Fp	i027_c	Ca3 Mn O17 S2
0.069	0.005	0.017	as input	P-62c	0.49	i027_d	Ca4 O17 S2
0.069	0.004	0.019	as input	P6(3)22	0.42	i027_e	Ca4 O17 S2

This structure was published in space group $P\bar{6}c2$ by Barkley *et al.*, *Acta Cryst. E67* (2011) i47-i48, despite a CheckCIF warning that the structure is probably centrosymmetric! However shelxt indicates $P\bar{6}m2$, which is also wrong. In fact the shelxt solution is the correct structure (and hence the lowest R1 value) but in the wrong space group because an inversion center has been overlooked. The correct space group, $P6_3/mmc$, had a higher R1 because the Mn atom has been incorrectly assigned as Ca.

When the figures of merit are similar, the centrosymmetric space group should almost always be chosen, unless the chemistry dictates otherwise. In this case the Flack parameter does not help because the deposited data had been merged in the wrong Laue group ('no Fp').

Command line options

If SHELXT started without a filename, it prints a list of possible commands.

–L is useful to override the Laue group specified by the SYMM cards, e.g. –L15 to try all trigonal and hexagonal space groups. –L16 converts orthorhombic to monoclinic with the current **a** axis made into the monoclinic **b**, and –L17 turns the current **c** into the unique monoclinic **b** axis.

For very tightly packed structures (e.g. some borides) the volume per atom should be reduced (e.g. –v8 rather than the default –v13).

If the default settings fail, try –y and –a (or even –a0.5) if the CC is good but the solution is a mess. If the CC values are all less than about 0.87, try –m1000. Also worth a try is varying –d in small resolution steps to cut out high resolution noise.

Limitations of SHELXT

The current version obtained fully correct structures for about 50% of the structures tested and most of the rest have the right atoms but some of them are wrongly assigned (typically N/C and S/P). The space group is found correctly in at least 97% of cases, although some of them had defeated XPREP. SHELXT showed that several of the ca. 700 test structures had been refined in the wrong space group!

Most of the small number of failures involved severe disorder, pseudo-symmetry or twinning (is $\langle |E^2-1| \rangle$ low?), violating the assumption that the structure consists of **atoms**. Non-centrosymmetric structures with a centrosymmetric arrangement of very heavy atoms can also be problematic. For equal-atom structures, data to 1.2Å are still required, though this may be relaxed when heavy atoms are present.

The largest structure solved with SHELXT had about 1400 unique non-hydrogen atoms in $P\bar{1}$, but the largest solved and assigned completely correctly had only 360 unique atoms in $P2_1/n$. These were particularly favorable cases. Usually SHELXD is more effective than SHELXT for very large or twinned structures.

Future plans

It is to be expected that SHELXT will make SHELXS obsolete but not SHELXD, which is still better for twins and very large equal-atom structures, and for locating the marker atoms in macromolecular phasing.

It would of course be possible to do more iterative optimization of the resulting structure than just one round of isotropic refinement. However the automatic structure determination in the Bruker APEX3 system already extends (SHEL)XT very effectively in this way.

Future plans include a better treatment of pseudo-symmetry (possibly with the help of the Friedel differences) and improved element assignments, as well as the automatic solution of twinned structures. This may take some time.

Acknowledgements and references

I am very grateful to all ca. 200 beta-testers for suggesting improvements, reporting bugs and providing interesting test data.

SHELXT: Sheldrick, G.M. (2015). *Acta Cryst.* **A71**, 3-8.

SHELXL: Sheldrick, G.M. (2015). *Acta Cryst.* **C71**, 3-8.

SADABS: Krause, L., Herbst-Irmer, R. Sheldrick, G.M. & Stalke, D. (2015). *J. Appl. Cryst.* **48**, 3-10.